

Investigation of the Electrochemical Properties of 3D Recyclable Aluminium doped LiMn₂O₄ Electrode: Improved Power and Energy Densities for Lithium-ion Battery Usage

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ABSTRACT

The Composites of Al-Doped LiMn₂O₄; Al_{0.1}:(LiMn₂O₄)_{0.9} and Al_{0.3}:(LiMn₂O₄)_{0.7}, were prepared using the hydrothermal method and drop casting deposition technique. The electrochemical performance of the Al-doped LiMn₂O₄ composite as a promising anode material for lithium-ion batteries was characterised by cyclic voltammetry analysis, electrochemical impedance spectroscopy and galvanostatic charge discharge analysis. The anodes' material exhibits a reversible capacity loss, which can be primarily linked to reverse reactions within the solid electrolyte interface formation, aluminium adsorption in the conducting LiMn₂O₄, and the electrolyte's electrochemical breakdown. The charges that are retained in the anode material during charging showed a linear decline in charge capacity as charging current intensity increased. Ionic polarisation was the reason for the observed drop in the charge and discharge capabilities at the current density of 5 A/g. Having greater specific capacitance and energy density, the composite Al_{0.1}:(LiMn₂O₄)_{0.9}, is a better anode material for electrochemical applications compared to Al_{0.3}:(LiMn₂O₄)_{0.7}, also its comparatively higher power density at a scan rate of 5 mV/s is mostly explained by its lower equivalent series resistance.

Keywords:

Power Density,
Energy Density,
Battery,
Anode,
Electrode.

INTRODUCTION

Lithium-ion battery (LIB) technology is thought to be in development. However, they must perform better for contemporary applications in order to achieve high coulombic efficiency, increase power and energy densities. The materials used and the production processes have a significant impact on the ultimate cost of a LIB (Ali *et al.*, 2022; Adesina, *et al.*, 2021; Kumar, *et al.*, 2019; Shichen *et al.*, 2023). It is necessary to note that in this industrial period there has seen a sharp rise in the need for energy. Due to their low cost, light weight, and ease of storage, fossil fuels; first carbon, then oil, and finally natural gas have been systematically exploited to power cities and factories (Chen *et al.*, 2019; Ahmad *et al.*, 2016; Zhao *et al.*, 2020; Zhang, *et al.*, 2019). But these sources also release a lot of toxic chemicals, such as carbon monoxide, sulphur dioxide, and different nitrogen oxides, making them unfriendly to the environment (Wang *et al.*, 2020). The energy sector has seen new trends and opportunities brought about by the lithium-ion battery, particularly in the burgeoning electric car market. Lithium-ion batteries need to be continuously improved in terms of capacity,

power output, life cycle, and safety in order to meet the demands of such applications (Yan *et al.*, 2020; Singh *et al.*, 2021; Zhang *et al.*, 2019). One way to improve these characteristics is to create a more effective and excellent anode electrode material (Diantoro *et al.*, 2024; Zhang *et al.*, 2020).

This work is aimed at synthesising and investigating the electrochemical properties of aluminium doped LiMn₂O₄ as an alternative anode material in LIBs which could be more sustainable and have reduced production costs compare to other graphitic materials.

The specific capacitance (C_{sp}) was evaluated from the equation:

$$C_{sp} = \frac{S}{2mk(E)} \quad (1)$$

S in (mA.V); m in (g); K in *mv/s*; and E in (V).

According to Alpha *et al.* (2019) and Zhang, *et al.* (2020), the energy density (E_D) and power density (P_D) were calculated using Equations (2) and (3)

$$E_D = \frac{1}{4} C_{sp} V^2 \quad (2)$$

$$P_D = \frac{V^2}{4 \times (ESR)m} \quad (3)$$

MATERIALS AND METHODS

Materials

The chemicals used in this study were of analytical grade with a percentage purity in the range of 99.892 %

- 99.999 %. Presented in Table 1 are the lists of chemicals used as precursor materials

Table 1: List of chemicals used as precursor materials

Chemicals	Chemical formula	Manufacturer (Company)	Purity
Lithium manganese IV oxide	LiMn_2O_4	Merck	99.995 %
Aluminium Carbonate	$\text{Al}_2(\text{CO}_3)_3$	Merck	99.995 %
Polyvinylidene fluoride	$-(\text{CH}_2=\text{CF}_2)_n-$	Merck	99.892 %
Ethanol	$\text{CH}_3\text{CH}_2\text{OH}$	Merck	99.995 %
N-methyl-2-pyrrolidone (NMP)	$\text{C}_5\text{H}_9\text{NO}$	Merck	99.999 %
De-ionized water	H_2O	-	99.995 %
Distilled water	H_2O	-	99.995 %

Method

Synthesis Al doped LiMn_2O_4 Composites

20 g of the LiMn_2O_4 powder, 0.1 mol/L aluminium carbonate solution were added to 1000 ml of distilled water and then stirred using a magnetic stirrer for 3 hours to obtain a composite, $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$. The composite was then filtered, after which the mixture was sonicated at 78.8 °C for 3 hours. After sonication, the sample was then washed with ethanol and dried in an electrical oven at 75 °C for 3 hours. The composite, $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$, was then grinded into a fine powder using a piston and mortar. Also, another 20 g of LiMn_2O_4 powder, 0.3 mol/L aluminium carbonate solution were added to 1000 ml of distilled water, and then stirred using a magnetic stirrer for 3 hours to obtain a composite, $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$. The composite was then filtered, after which the mixture was sonicated at 78.8 °C for 3 hours. After sonication, the composite was then washed with ethanol and dried in an electrical oven at 75 °C for 3 hours. The composite, $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$,

was then grinded into a fine powder using a piston and mortar.

Preparation of Composites for Characterisation

In the preparation of the composite material for characterisation, a stainless-steel substrate was used. The substrate was first smoothed after which it was washed with distilled water. The substrate was then soaked in an acetone for 1 hour before washing again with distilled water, after which it was ultrasonicated for 30 minutes and then dried in an electric oven for 1 hour. The electrode material was prepared using the drop casting method of deposition. 90 % sample provide, 10 % polyvinylidene fluoride (PVDF) binder were dissolved in N-methyl-2-pyrrolidone (NMP) solvent and stirred for 5 hours until it formed slurry. It was then loaded uniformly onto the treated stainless-steel substrate. After deposition, it was then dried in an electric oven at 120 °C for 5 hours. The electrode is formed and ready for characterisation.

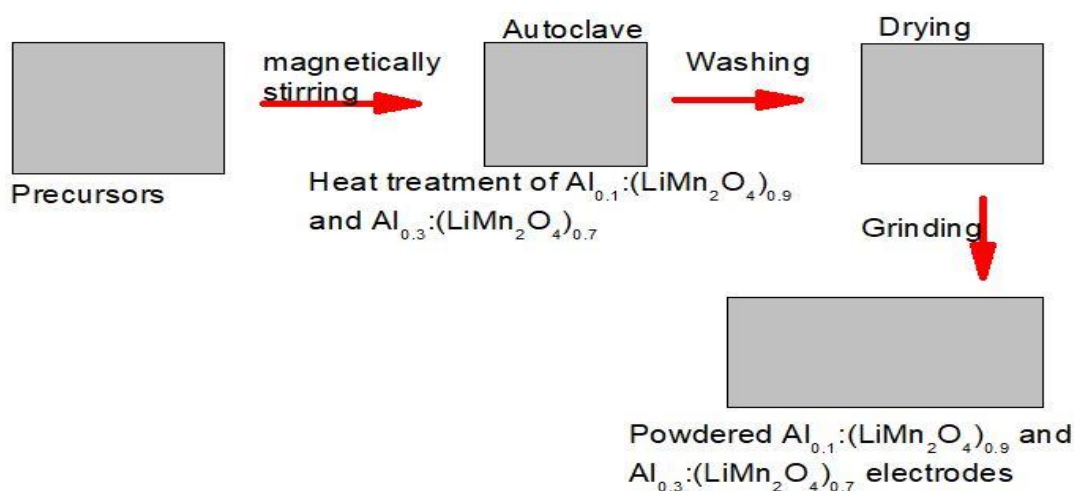


Figure 1: Schematic diagram for the hydrothermal synthesis of $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ and $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$

The electrochemical analysis was carried out in a three electrode configuration with the active material serving as the working electrode, Ag/AgCl serving as the reference through which current flows into the system at a potential window voltage of 0.9 V at a 5 mV/s scan rate and a glassy carbon rod serving as the counter electrode through which the current flows out. A 2 M KOH aqueous solution served as the electrolyte, which provided a medium for current flow and ion interaction, using the electrochemical analyser model CH1604E.

RESULTS AND DISCUSSION

Structural Analysis

X-ray Diffraction (XRD) Analysis for $Al_{0.1}:(LiMn_2O_4)_{0.9}$ and $Al_{0.3}:(LiMn_2O_4)_{0.7}$

Figure 2 shows the stacked crystallites of the XRD pattern for $Al_{0.1}:(LiMn_2O_4)_{0.9}$ and $Al_{0.3}:(LiMn_2O_4)_{0.7}$. This shows the most preferred orientation to be at the diffraction plane (101) (111), and (100) at Bragg's angles of 27.20° , 25.29° , and 23.26° , respectively both composites. This indicated the presence of the crystalline structure of the Al^{3+} and Li^+ in the network of the Mn_2O_4 . At higher Bragg's diffraction angles between 30° and 70° , the composites are almost completely amorphous impurities.

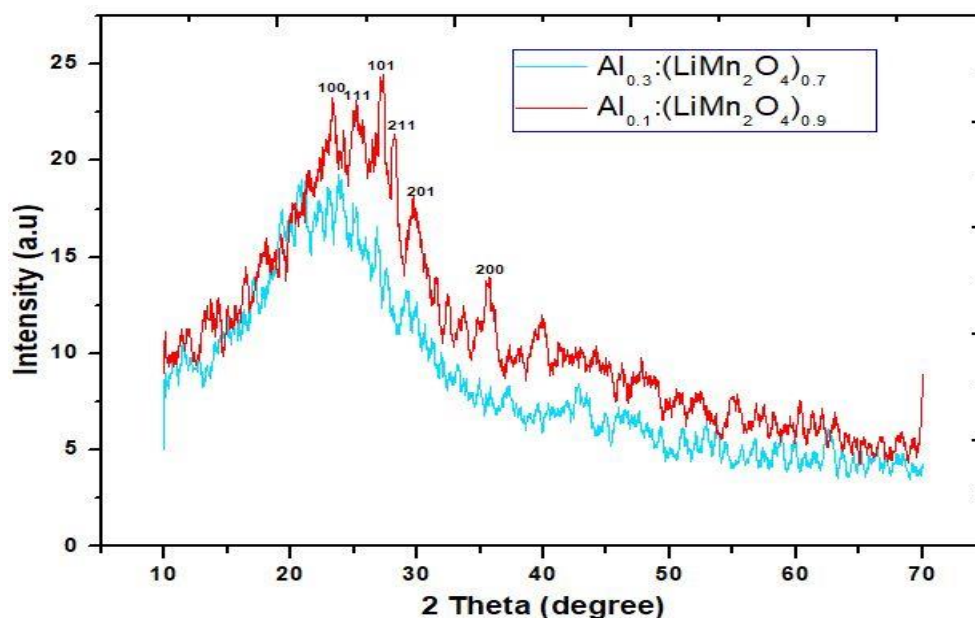
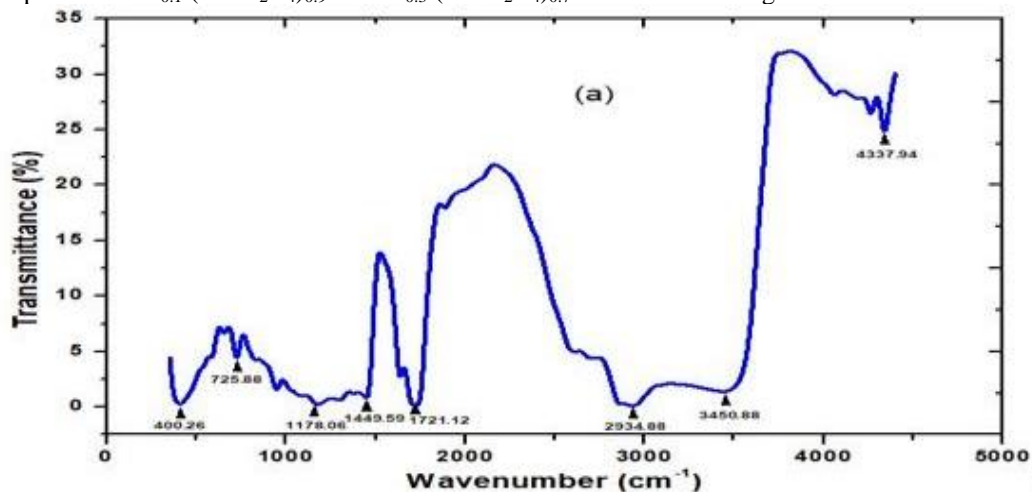


Figure 2: XRD pattern for $Al_{0.1}:(LiMn_2O_4)_{0.9}$ and $Al_{0.3}:(LiMn_2O_4)_{0.7}$

FTIR Spectra Analysis for $Al_{0.1}:(LiMn_2O_4)_{0.9}$ and $Al_{0.3}:(LiMn_2O_4)_{0.7}$

The FTIR spectra for $Al_{0.1}:(LiMn_2O_4)_{0.9}$ and $Al_{0.3}:(LiMn_2O_4)_{0.7}$ were shown in Figure 3.



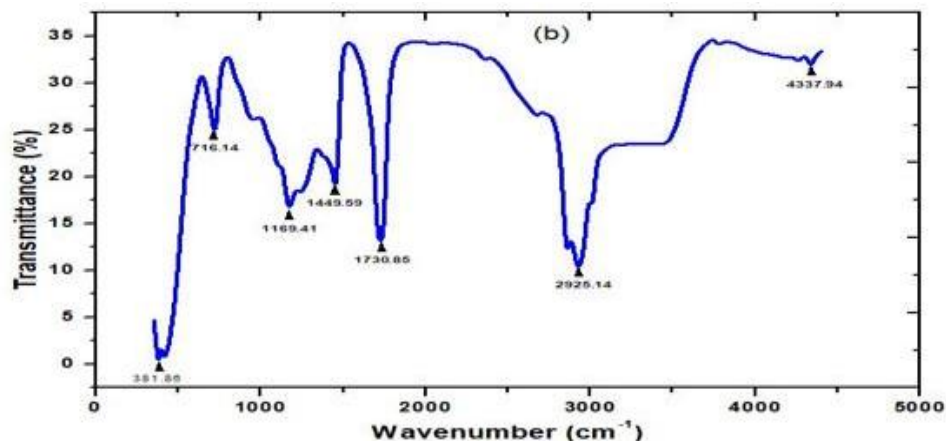


Figure 3: FTIR for (a) $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ and (b) $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$

The aluminium oxygen functional group was detected in the spectra for both composites between 400 and 1721.12 cm^{-1} , indicating absorption brought on by the bond molecules. This revealed the features of the oxygen molecules' sp^3 vibration, which fall within the group frequency range. At the characteristic peaks, 2934.88 cm^{-1} and 2926.14 cm^{-1} , for $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ and $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$, respectively, the aluminium oxygen ionic bond was typical at the characteristic peaks, 2934.88 cm^{-1} and 2926.14 cm^{-1} . The structural distortion of the LiMn_2O_4 may have been the cause of this change. This might have resulted from the Al ion being introduced into the LiMn_2O_4 structure's inter-planner space, which changed the bond distances between the Al, Mn, O, and Li atoms. This structural deformation enhances ionic transport during the electrochemical process.

Electrochemical Analysis

Figure 3 shows the cyclic voltammograms from the cyclic voltammetry analysis for $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ and $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$ respectively at scan rates of 3 and 5 mVs^{-1} , current density of 100 mA/g . Figures 4 and 5 give the plots for the GCD at 5 A/g for $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$, and $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$. Figures 6 and 7 provide the Nyquist plots from the Electrochemical Impedance Spectroscopy analysis for $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$, and $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$ composites.

Cyclic Voltammetry Analysis

The values of C_{sp} , E_{D} and P_{D} were computed using Equations 1, 2, and 3. The summary of the results from the electrochemical analysis for the $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ and $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$ composite at scan rates of 3 and 5 mVs^{-1} are given in Tables 2 and 3.

Table 2: The Values of C_{SP} and E_{D} and P_{D} @ 3mV/s

Composite	Mass (g)	Potential (V)	ERS (Ω)	C_{sp} (F/g)	E_{D} (Wh/kg)	P_{D} (kW/kg)
$\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$	0.119	1.4	2.0	847.99	115.40	2.059
$\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$	0.119	1.4	2.5	484.80	66.00	1.647

Table 3: The Values of C_{SP} and E_{D} and P_{D} @ 5mV/s

Composite	Mass (g)	Potential (V)	ERS (Ω)	C_{sp} (F/g)	E_{D} (Wh/kg)	P_{D} (kW/kg)
$\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$	0.119	1.4	2.0	508.799	53.075	0.349
$\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$	0.119	1.4	2.5	353.57	36.7630	0.349

Shown in Figure 4 are the CV curves of the samples acquired at scan rates of 3 and 5 mVs^{-1} between 0.0 and 1.4 V potential window. The insertion or extraction of aluminium from the K LiMn_2O_4 lattice was shown to correlate to only one pair of redox peaks. Aluminium is also extremely reversible when it comes to its insertion or extraction from the LiMn_2O_4 composite because the CV curves for the cycles are nearly identical. The anode material exhibits a significant reversible capacity loss, which is primarily explained by side reactions including solid electrolyte interface (SEI) formation, aluminium

adsorption, and the electrochemical electrolyte breakdown. The composite $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ at a scan rate of 3 mV/s gives a higher specific capacitance, energy density, and power density, thereby making it a more suitable anode material when used in battery applications this agrees with *alpha et al.* (2019). The composite $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ at a scan rate of 5 mV/s still maintains a relatively higher power density, this is mainly attributed to its lower equivalent series resistance.

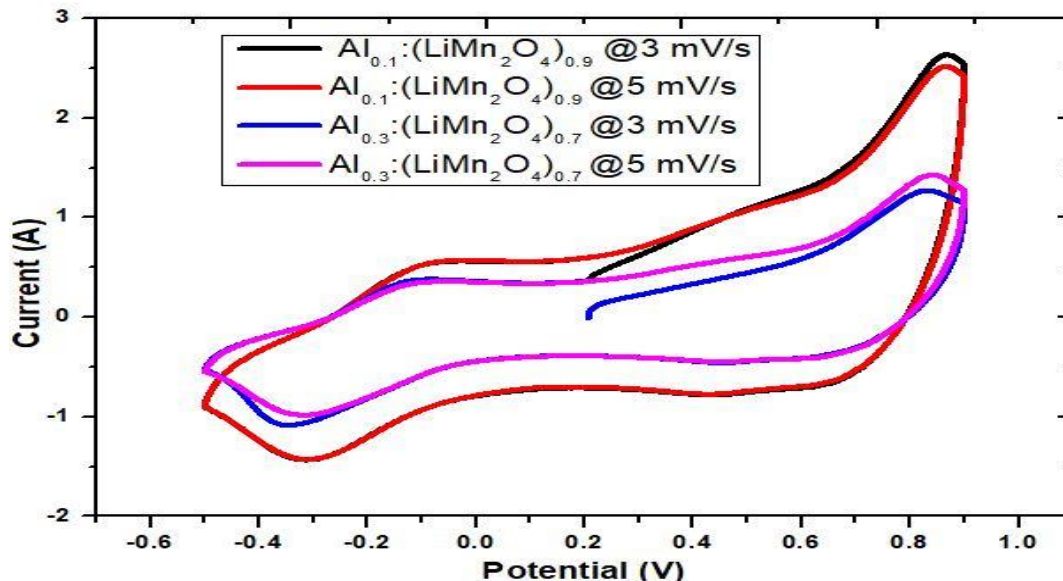


Figure 4: Cyclic Voltammogram curve at different scan rates for $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$ and $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ composites

GCD Characterisation

The cyclic voltammetry results, which show that redox currents are only set up in this voltage range with an oxidation peak around 0.9 V, demonstrate that the primary cause of the change in the charging voltage is Al^{3+} insertion (the voltage decreases) and extraction (the voltage increases) into and from the LiMn_2O_4 anode. High currents indicate rapid changes in the Al ion concentration within the LiMn_2O_4 structure's lamellar

planes, which in turn implies quick changes in the anode potential over a short amount of time. The values of the specific capacitance and energy densities for $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ and $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$ respectively at current density of 5 A/g for the GCD are shown in Tables 4 and 5. The charge stored in the anode material during charging is represented by a linear reduction in charge capacity with increasing charging current intensity, as seen in Figures 5 and 6.

Table 4: Values of C_{SP} and E_D for $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ @5A/g and 1.4 V for GCD

	First cycle	Second cycle	Third cycle	Fourth cycle	Fifth cycle
ΔT	69.15	68.97	68.14	67.91	66.72
ΔV	1.4	1.4	1.4	1.4	1.4
C_{SP}	246.96	246.32	243.34	242.53	238.28
E_D	33.613	33.526	33.121	33.011	32.433

Table 5: Values of C_{SP} and E_D for $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$ @5A/g and 1.4 V for GCD

	First cycle	Second cycle	Third cycle	Fourth cycle	Fifth cycle
ΔT	23.27	22.39	22.01	21.47	21.39
ΔV	1.4	1.4	1.4	1.4	1.4
C_{SP}	83.107	79.961	78.607	76.678	76.392
E_D	11.311	10.883	10.699	11.436	10.397

The capacitance retention is seen in Figures 6 and 7 for $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ and $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$, having capacity retention of 96.41 % and 91.92 % after five cycles at 5 A/g, respectively.

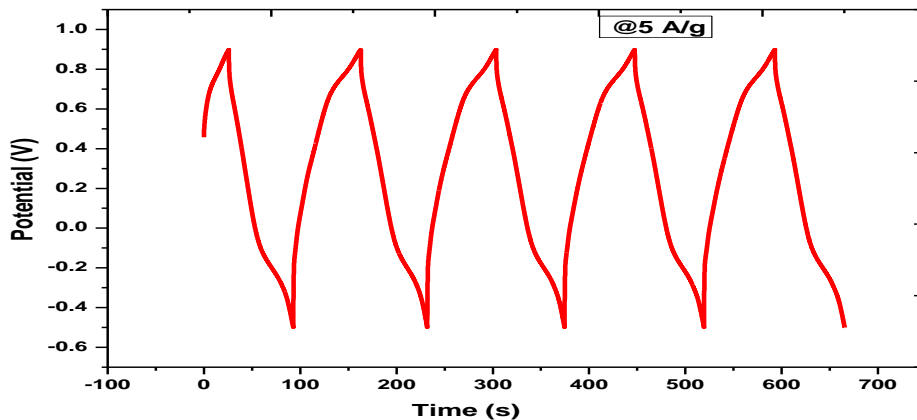


Figure 5: Shows the galvanostatic charge discharge GCD pattern for $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ composites @ 5A/g

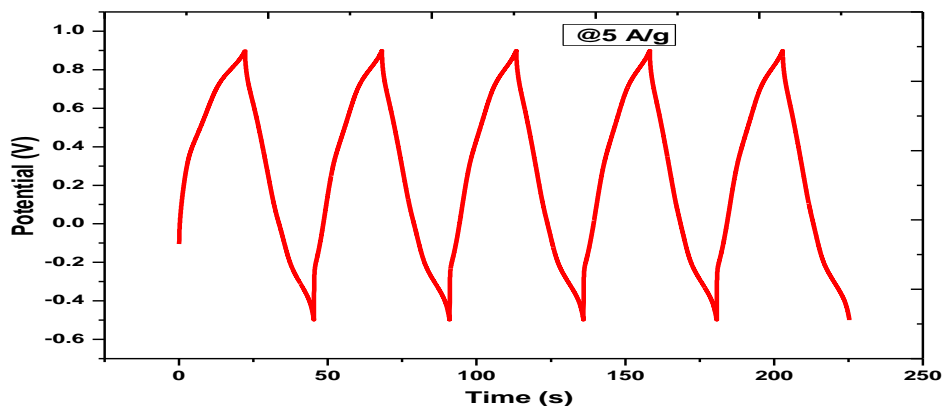


Figure 6: Shows the galvanostatic charge discharge GCD pattern for sample $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$ composites @ 5 A/g

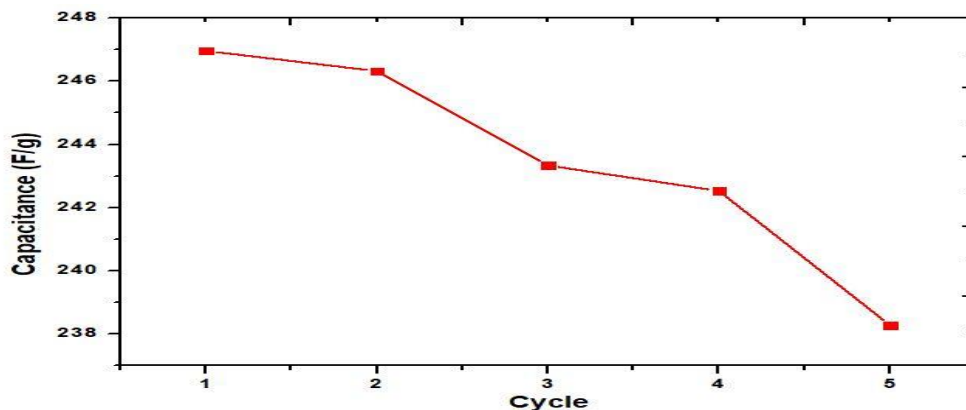


Figure 7: Capacity retention for $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ composites @ 5 A/g

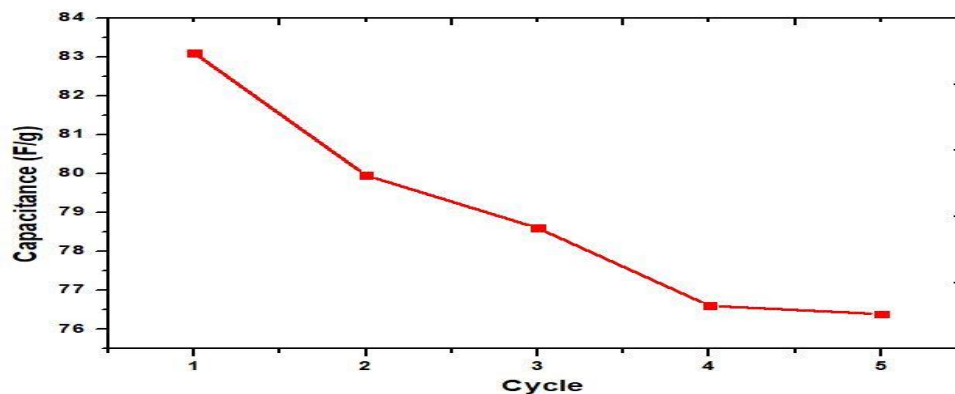


Figure 8: Capacity retention for Al_{0.3}:(LiMn₂O₄)_{0.7} composites @5 A/g

EIS Characterisation

In finding out more about the electrochemical properties of the composite anode electrode, EIS measurement was carried out in the frequency range of

100,000 to 0.1 Hz from which the equivalent series resistance (ERS) were estimated. This is seen in Figures 9 and 10.

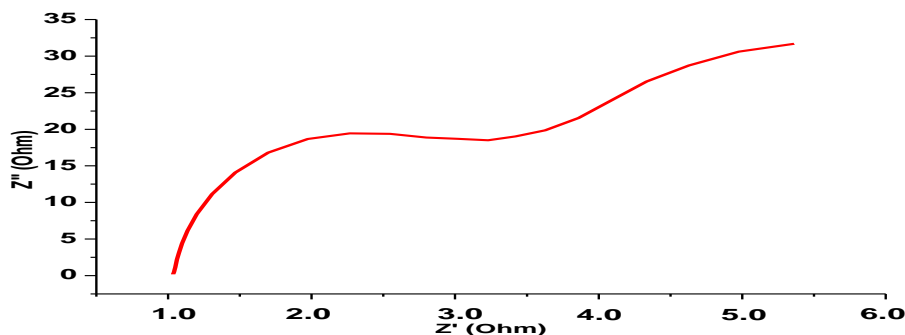


Figure 9: Electrochemical Impedance Spectroscopy Analysis Showing the Nyquist Plot for Al_{0.1}:(LiMn₂O₄)_{0.9}

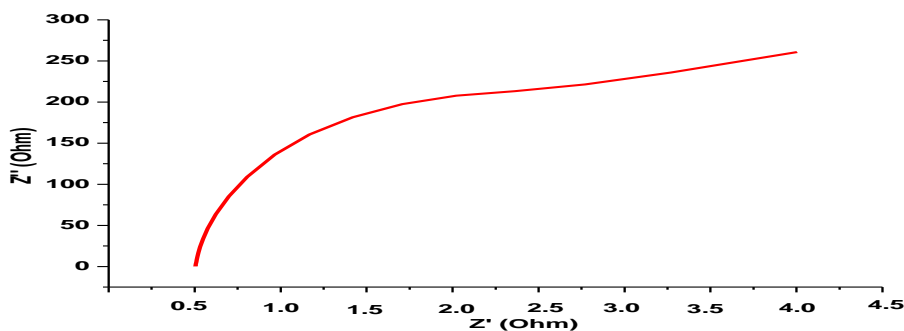


Figure 10: Electrochemical Impedance Spectroscopy Analysis Showing the Nyquist Plot for Al_{0.3}:(LiMn₂O₄)_{0.7}

CONCLUSION

Al³⁺ was added to the LiMn₂O₄ network to enhance charge transfer and ion transport kinetics throughout the composite electrodes. Through the intercalation of cations within the composite material's network, the electrode materials exploited the synergistic advantage. The anodes' material exhibits a reversible capacity

loss, which can be primarily linked to reverse reactions within the solid electrolyte interface formation, aluminium adsorption, and the electrolyte's electrochemical breakdown. The charge that is retained in the anode material during charging shows a linear decline in charge capacity as the charging current intensity increases. The observed decrease in charge

and discharge capabilities at the current density of 5 A/g can be explained by ionic polarisation. With a greater specific capacitance and energy density, the composite $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.9}$ is a better anode material for electrochemical applications compared to $\text{Al}_{0.3}:(\text{LiMn}_2\text{O}_4)_{0.7}$. The comparatively higher power density of $\text{Al}_{0.1}:(\text{LiMn}_2\text{O}_4)_{0.9}$ composite at a scan rate of 5 mV/s is mostly explained by its lower equivalent series resistance.

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